VIBRATIONAL AND THERMODYNAMIC PROPERTIES OF 1,3,5-TRIAMINO-2,4,6-TRINITROBENZENE (TATB): COMPARISON OF EXCHANGE-CORRELATION FUNCTIONALS IN DENSITY FUNCTIONAL THEORY

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Vibrational and thermodynamic properties of TATB have been investigated within the quasiharmonic approximation and density functional theory using three exchange-correlation functionals: local density approximation (LDA), generalized gradient approximation (GGA), and GGA with an empirical van der Waals correction (GGA + vdW). We find that GGA provides a reasonable description of only the heat capacity and thermal expansion, while it fails to reproduce the experimental bulk modulus and volume. Van der Waals correction improves the lattice constants, volume, and bulk modulus, but it fails badly in describing thermal expansion and heat capacity. In contrast, LDA accurately describes all the thermodynamic properties of TATB considered here. For example, the equilibrium volume calculated with LDA is only 4.6% smaller than the experimental value after including vibrational contributions. It is therefore essential to include phonon contributions when comparing the calculated volume with experimental data at ambient conditions. We show that an accurate equation of state of TATB is obtained by simply multiplying the volume calculated with LDA by a factor of 1.046, because LDA predicts the bulk modulus well in the entire pressure range. Therefore, LDA is a satisfactory exchange-correlation functional for TATB because only LDA correctly predicts the volume dependence of vibrational frequencies. All calculations exhibit an abrupt change of the compressibility at a critical pressure, $P_c \sim 0.5$ –1.0 GPa. Below P_c , the volume reduction by pressure is mainly due to the lattice contraction along the c axis, whereas above P_c the lattice contracts significantly along all three axes.

KEY WORDS: TATB, Vibrational and thermodynamic properties, DFT Simulations

1. INTRODUCTION

1,3,5-Triamino-2,4,6-trinitrobenzene (TATB) is an important molecular crystal widely used in high-performance energetic applications. Its remarkable stability under thermal, impact, and shock-initiation conditions (Dobratz, 1995; Travis, 1992), as well as its nonlinear optical properties (Ledoux et al., 1990; Son et al., 1999; VoigtMartin et al., 1996; VoigtMartin et al., 1997), have stimulated numerous studies (Makashir and Kurian, 1996; Becuwe and Delclos, 1993; Osmont et al., 2007; Agrawal, 2005; Boddu et al., 2010; Bourasseau et al., 2011; Boyer and Kuo, 2007; Budzevich et al., 2010; Byrd and Rice, 2007; Cady and Larson, 1965; De Lucia et al., 2003; Deopura and Gupta, 1971; Dobratz, 1995; Fedorov and Zhuravley, 2014; Fell et al., 1998; Filippini and Gavezzotti, 1994; Fried and Ruggiero, 1994; Grebenkin and Kutepov, 2000; Huang et al., 2014; Kolb and Rizzo, 1979; Kroonblawd and Sewell, 2013; Kunz, 1996; Liu et al., 2011; Losada and Chaudhuri, 2009, 2010; Lurnan et al., 2007; Chevalier et al., 1993; Talawar et al., 2006; Mang and Hjelm, 2013; Margetis et al., 2002; McGrane et al., 2005; McGrane and Shreve, 2003; Olinger and Cady, 1976; Pravica et al., 2009; Price, 1988; Qian et al., 2014; Toghiani et al., 2008; Rosen and Dickinson, 1969; Satija et al., 1991; Sewell, 1997; Shan et al., 2014; Son et al., 1999; Stevens et al., 2008; Willey et al., 2009; Taylor, 2013; Thadhani, 1994; Towns, 1983; Traver, 2010; Vergoten et al., 1985; Wu and Fried, 2000; Yetter et al., 1995; Huang et al., 2005; Zeman, 1993; Zhang et al., 2012a, 2012b, 2009). For example, these studies measured thermodynamic properties such as thermal expansion at ambient pressure (Kolb and Rizzo, 1979) and equation of state at room temperature (Kolb and Rizzo, 1979; Stevens et al., 2008), and vibrational properties such as Raman and infrared spectroscopy (Deopura and Gupta, 1971; Satija et al., 1991; Towns, 1983; Vergoten et al., 1985). Although the stability of TATB is usually attributed to its significant inter- and intramolecular hydrogen bonding (Dobratz, 1995), the underlying atomistic mechanisms are still unclear. Detailed structural, vibrational, and thermodynamic properties are indispensable for understanding the atomistic origin of the unusual properties of TATB crystal.

Extensive theoretical studies on TATB crystal have been conducted using empirical interatomic potentials (Bedrov et al., 2009; Gee et al., 2004; Pastine and Bernecker, 1974; Rai et al., 2008; Sewell, 1996) or electronic-structure calculations (Byrd et al., 2007; Grebenkin and Kutepov, 2000; Kunz, 1996; Liu et al., 2007; Budzevich et al., 2009; Oleynik et al., 2007; Wu and Fried, 2000; Zhu et al., 2009) based on the density functional theory (DFT) and Hartree-Fock method. The interatomic potentials describe the crystal structure and bulk modulus at ambient conditions accurately because their parameters are fitted to reproduce experimental data. Furthermore, these potentials have been shown to predict TATB properties at high pressures and temperatures reasonably well (Bedrov et al., 2009; Gee et al., 2004; Rai et al., 2008). In contrast, nonempirical electronic-structure calculations usually have considerable errors in crystal properties of TATB (Byrd et al., 2007; Liu et al., 2007; Wu et al., 2003) and other molecular crystals (Byrd et al., 2007; Byrd et al., 2004; Conroy et al., 2008a, 2008b; Qiu et al., 2006;

Shimojo et al., 2010; Wu et al., 2011; Zhao and Liu, 2008). For example, the generalized gradient approximation (GGA), which is a commonly used exchange-correlation (xc) functional in DFT calculations, tends to overestimate the equilibrium volume of TATB by $20 \sim 30\%$ (Byrd et al., 2007; Liu et al., 2007), while another popular xc functional, the local density approximation (LDA), underestimates it by $10 \sim 15\%$ (Byrd et al., 2007; Liu et al., 2007; Wu et al., 2003). Such failure of DFT is often ascribed to its inadequacy in describing the dispersion interaction. As a result, there is great interest in improving GGA calculations by incorporating van der Waals (vdW) corrections (Dion et al., 2004; Grimme, 2004; Shimojo et al., 2010; Tkatchenko et al., 2010; Tkatchenko and Scheffler, 2009). Another essential issue is the effect of phonons on the thermodynamic properties of molecular crystals. Although the phonon effect on the equilibrium volume is small (about 1-2%) for crystals with larger bulk moduli (Karki et al., 2000; Li et al., 2007; Mounet and Marzari, 2005; Pavone et al., 1993; Sun et al., 2008; Wentzcovitch et al., 2010; Wu and Wentzcovitch, 2007; Wu et al., 2008; Yu and Wentzcovitch, 2006), this effect can be very large for energetic crystals, which exhibit large thermal expansion.

Recently, we reported a detailed investigation of vibrational and thermodynamic properties of the β -HMX (octahydro-1,3,5,7-tetranitro-1,3,5,7-tetrazocine) crystal with LDA, GGA, and GGA with an empirical van der Waals correction (GGA+vdW) (Matsui et al., 2012). We found that phonons significantly increase the equilibrium volume (by more than 5%). LDA predicts the ambient volume to be only 6% less than the experimental value after including the vibration contribution. Most importantly, LDA accurately predicts other thermodynamic properties such as the bulk modulus, heat capacity, and thermal expansion. In contrast, GGA only predicts the heat capacity and thermal expansion well, and produces large errors in the bulk modulus and volume. Although an empirical vdW correction to GGA was found to improve the volume and bulk modulus, the GGA+vdW calculation fails badly in predicting the thermal expansion and heat capacity. Therefore, contrary to the common belief, LDA is a rather good xc functional for β -HMX. In order to clarify whether these observations also hold for other energetic crystals, here we study the vibrational and thermodynamic properties of the TATB crystal in the framework of DFT with three xc functionals—LDA, GGA, and GGA+vdW—to identify the best functional. The other main purpose of this paper is to provide detailed thermodynamic properties of TATB at high temperatures and pressures based on DFT with the best functional thus identified, which are not available in literature to the best of our knowledge.

2. CALCULATION DETAILS

The calculations are performed using pwscf and phonon programs of the Quantum Espresso software for DFT calculations based on the plane-wave basis and pseudopotentials (Giannozzi et al., 2009). The plane-wave cutoff energy (E_{cut}) of 30 Ry is found sufficient to obtain the converged results. Brillouin-zone summations over electronic

states are performed over a $2 \times 2 \times 2$ (2 points) k mesh with a (1/2, 1/2, 1/2) shift from the origin. The Vanderbilt ultrasoft pseudopotentials are generated for C, N, H, and O (Vanderbilt, 1990). For the exchange-correlation functional, we use LDA (Ceperley and Alder, 1980; Perdew et al., 1981) and GGA by Perdew, Burke, and Ernzerhof (PBE) (Perdew et al., 1996).

Van der Waals correction is included using Grimme's DFT-D2 methodology (Grimme, 2004, 2006). The empirical dispersion correction for the energy is given by

$$E_{disp} = -s_6 \sum_{i < j} \frac{C_{ij}}{R_{ij}^6} f_{damp}(R_{ij}) \tag{1}$$

where C_{ij} and R_{ij} denote the dispersion coefficient and the interatomic distance between the *i*th and *j*th atoms, and s_6 is a global scaling factor that only depends on the density functional used. A damping function f_{damp} is introduced to ensure that the dispersion correction is negligible for small R_{ij} .

Structural optimizations, with and without the vdW correction, are achieved using the damped variable cell shape molecular dynamics method (Wentzcovitch, 1991). For each fully optimized structure, dynamical matrices are computed on a $2 \times 2 \times 2$ q (wave vector) mesh based on the density functional perturbation theory (DFPT). The dynamical matrices (Baroni et al., 2001) are modified by incorporating the vdW contribution if the structure has been optimized with the vdW correction. The results are then interpolated in a regular $8 \times 8 \times 8$ q mesh to obtain the vibrational density of states.

The Helmholtz free energy in the quasiharmonic approximation (QHA) is given by

$$F(V,T) = U_0(V) + \frac{1}{2} \sum_{q,j} h \varpi_j(q,V) + k_B T \sum_{q,j} \ln\{1 - \exp[-h\varpi_j(q,V)/k_B T]\}$$
(2)

where q is a wave vector in the first Brillouin zone, j is an index of the phonon mode with frequency $\varpi_j(q, V)$, V and T are the volume and temperature of the system, and k_B and h are the Boltzmann and Planck constants. The first, second, and third terms in Eq. (2) are the static internal, zero-point, and vibrational energy contributions, respectively. The calculated Helmholtz free energy versus volume is fitted by the Birch-Murnaghan thirdorder finite strain equation of states, which in turn is used to derive thermodynamic properties.

3. RESULTS AND DISCUSSION

3.1 Lattice Structure

The structure of the TATB crystal is shown in Fig. 1. The unit cell of TATB is triclinic with space group $P\overline{1}$. Each unit cell has two TATB molecules. TATB crystal is highly anisotropic with planar molecules, whose planes are nearly parallel to the ab plane. The



FIG. 1: (a) TATB crystal structure projected along the [001] axis. For clarity of presentation, only the top molecules in the unit cells are shown, namely, for atoms with 0.5 < z < 1.0, where z is the coefficient for the basic vector along the c lattice vector. Intramolecular and intermolecular hydrogen bonds are displayed with green solid lines and dashed lines, respectively. (b) TATB crystal in the primitive cell. Carbon, nitrogen, oxygen, and hydrogen atoms are shown in black, blue, red, and green, respectively.

interaction between different planes is dominated by vdW interaction, while the intermolecular interaction within the ab plane mainly results from the hydrogen bonds (see green solid and dashed lines in Fig. 1). The anisotropy in TATB structure is clearly reflected in the errors of the lattice parameters by DFT calculations (see Table 1). The errors of a and b are far smaller than those of c for both GGA and LDA. LDA calculations predict a and b quite well but have a significant error in c. GGA calculations show significant errors in all directions but have a dominant error in c. These features on lattice parameters in the LDA and GGA calculation are consistent with previous calculations (Byrd et al., 2007; Liu et al., 2007). In particular, GGA static calculation overestimates the equilibrium volume by 27%, which clearly indicates that GGA is inadequate in treating vdW interaction. (Note that the interaction between different TATB planes along the c axis is dominated by vdW interaction as explained above.) The empirical vdW correction dramatically reduces the errors in the lattice parameters using GGA, which is consistent with previous works (Budzevich et al., 2010; Landerville et al., 2010). The equilibrium volume using GGA+vdW is only 0.9% smaller than the experimental value.

Budzevich et al had indicated anisotropic behavior of TATB under uniaxial compressions (Budzevich et al., 2010). The variations of the lattice parameters of the TATB

Volume 14, Number 6, 2015

TABLE 1: Lattice parameters $(a, b, c, \alpha, \beta, \gamma)$, volume (V), and bulk modulus (K_T) of TATB crystal at a pressure of 0 GPa. Both static (without phonon contributions) and QHA calculations within the framework of DFT using LDA, GGA, and GGA+vdW functionals are shown, along with the corresponding experimental data. The numbers in parentheses are deviations from the experimental value in percentage

a (Å)	b (Å)	c (Å)	α (°)	β (°)	γ (°)	V (Å ³)	K_T (GPa)	
8.864	8.881	6.373	108.67	92.17	120.00	399.6	10.7	IDA (statia)
(-1.6)	(-1.6)	(-6.4)	(0.1)	(0.4)	(0.0)	(-9.6)	19.7	LDA (static)
8.939	8.954	6.641	107.8	92.08	120.0	422.3	11.9	LDA (300
(-0.7)	(-0.8)	(-2.5)	(-0.7)	(0.3)	(0.0)	(-4.6)	11.0	K)
9.307	9.324	7.894	105.10	92.29	120.11	560.1	2.4	GGA
(3.3)	(3.3)	(15.9)	(-3.2)	(0.5)	(0.1)	(26.6)	2.4	(static)
9.346	9.361	8.226	104.30	92.27	120.14	591.6	1 29	GGA*
(3.7)	(3.7)	(20.8)	(-4.0)	(0.5)	(0.1)	(33.7)	1.30	(300 K)
9.103	9.131	6.634	109.07	92.11	119.92	438.3	17.1	GGA+vdW
(1.0)	(1.1)	(-2.6)	(0.4)	(0.3)	(0.0)	(-0.9)	17.1	(static)
9.138	9.162	6.740	108.81	92.06	119.94	449.6	12.86	GGA+vdW
(1.4)	(1.4)	(-1.0)	(0.2)	(0.2)	(0.0)	(1.6)	12.00	(300 K)
							$13.6 \sim 16.7$	Expt.
0.010	0.028	6 8 1 2	108 50	01.82	110.07	112 5	(Olinger and Cady,	(Olinger
9.010	9.020	0.012	100.39	91.02	119.97	442.3	1976; Stevens et	and Cady,
							al., 2008)	1976)

^{*}The effect of zero-point motion to the volume is ignored in order to avoid the extrapolation.

crystal under pressure also exhibit a large anisotropy. LDA, GGA, and GGA+vdW calculations all show the same feature that the anisotropy decreases abruptly at a critical pressure. Results from the GGA calculation are shown in Fig. 2. Note that a and b have nearly identical dependencies on the pressure, which is consistent with previous works (Budzevich et al., 2010). There is a sudden change in the compression ratio of a, b, and c, which is defined as $\Delta a/a:\Delta b/b:\Delta c/c$, at about 0.5 GPa (Δa , Δb , and Δc are the reduction of a, b, and c, respectively, compared with their zero-pressure values). Below 0.5 GPa, a and b change much more slowly with pressure than c. The compression ratio along the a, b, and c axes is about 1.0:0.94:9.7. The compression in c is one order of magnitude larger than that in a and b. Volume reduction by pressure thus arises mainly from the variation of c at low pressures. Above 0.5 GPa, a and b change much more rapidly with pressure than they do at low pressures. The compression ratio is about 1:1:1.95, which indicates that the variation of a and b also makes a significant contribution to volume reduction at pressures higher than 0.5 GPa. LDA calculation exhibits the same feature, though with a higher critical pressure, ~1.0 GPa. The compression



FIG. 2: The variation of (a) lattice constants a and b and (b) the lattice constant c under compression calculated with the GGA.

ratio calculated by LDA is 1:1.04:5.92 and 1:0.99:2.79, respectively, below and above 1.0 GPa.

3.2 Vibrational and Thermodynamic Properties

The vibrational properties of TATB crystal at zero pressure are listed in Table 2 for Raman modes and Table 3 for infrared modes. The dependence of the calculated vibrational properties on the xc functional is similar to that in β -HMX (Matsui et al., 2012). For phonon modes with frequency $>3000 \text{ cm}^{-1}$, the LDA predicts a smaller frequency than the GGA. But for the other modes except a few, frequencies from the LDA are always larger than those from the GGA. This is consistent with the fact that the LDA predicts the equilibrium volume to be 36% smaller than the GGA. Despite the large volume difference between the LDA and GGA calculations, the corresponding frequency difference is small for modes with frequency >1000 cm⁻¹. Both the LDA and GGA results agree well with the experimental data for these modes (Deopura and Gupta, 1971; Satija et al., 1991; Towns, 1983; Vergoten et al., 1985). However, for the modes with frequency $<300 \text{ cm}^{-1}$, the LDA and GGA results differ significantly, with the LDA results more consistent with the experimental data. The difference between the GGA and GGA+vdW calculations arises from: (1) the reduction of the equilibrium volume at 0 GPa by 28% due to the vdW effect; and (2) the vdW corrections on the force constants. These effects are negligible for high-frequency modes but are significant at low frequencies. The GGA+vdW results are more consistent with the experimental data than the GGA results.

The mode Grüneisen parameter $\gamma_{i,q}$ describes how the phonon frequency depends on the volume and is critical to the computation of thermodynamic properties. It is defined as

[ABLE 2: Raman modes of TATB crystal at 0 GPa calculated with the LDA, GGA, and GGA+vdW functionals, along with
he experimental data at ambient condition. γ is the mode Gruneisen parameter [see Eq. (3)] at the gamma point, and γ is the
iverage of the mode Grüneisen parameters in the first Brillouin zone [see Eq. (4)]

	1)	Satija et	al. (1991)																		
	periment (cm	Deopura and	Gupta (1971)																		1303
	ExJ	Vergoten et	al. (1985)	3315			3215			1598		1560			1438		1413	1372		1315	
			ح	0.02	0.01	0.01	0.02	0.02	0.01	0.05	0.05	0.04	0	-0.01	0	0.01	-0.02	-0.02	0.07	0	-0.01
		Wbv+A	٨	0.02	0.02	0.02	0.02	0.03	0.02	0.09	0.08	0.07	0	-0.02	0.02	0.02	-0.07	-0.04	0.09	0.05	0.03
		GG/	ω (cm ⁻¹)	3389.5	3386.9	3374.9	3277.3	3269.3	3259	1562.8	1553.2	1543	1497.5	1489.6	1440.1	1439.4	1394.6	1378.4	1349.8	1295.1	1288.4
			٦	0.01	0.01	0.01	0.01	0.01	0.01	0.03	0.02	0.01	0	0	0	0	0	-0.01	0.02	0	0
	culation	GGA	٨	0.01	0	0.01	0.01	0.01	0.01	0.02	0.02	0.01	0	0	0	0	-0.02	-0.03	0.02	0.02	0
in parameter	Cal		$\omega (cm^{-1})$	3374.3	3370.8	3362.4	3266.6	3257.3	3251.7	1550.6	1544.3	1536.7	1483.7	1480.6	1439.6	1438.9	1394.6	1381	1339.8	1286	1281.1
			4	0.02	0.01	0.01	0.02	0.02	0.01	0.08	0.08	0.07	0.03	0.02	0.02	0.01	0.01	0.02	0.04	0.05	0.05
		DA	7	0.02	0.01	0.01	0.02	0.02	0.01	0.08	0.07	0.06	0.02	0	0.03	0.03	0.01	0.07	-0.02	0.05	0.03
avoiugo oi u.		L	ω (cm ⁻¹)	3257.4	3253.7	3242.7	3167.9	3155.6	3146.8	1599.6	1588.8	1578.8	1484.7	1474.7	1460.5	1458.4	1414.2	1384.2	1346.1	1327.6	1318.2

International Journal of Energetic Materials and Chemical Propulsion

	(Satija et	al. (1991)					1170			1030		890			831							
	periment (cm $^{-1}$	Deopura and	Gupta (1971)	1229				1154	1129		1017			880				819					
	ExJ	Vergoten et	al. (1985)	1212			1188	1164	1137		1023			878			828						
			Ā	0.03	0.04	0.04	0.01	0.01	0.05	-0.08	-0.02	-0.01	-0.09	-0.1	0.07	0.04	0.01	0	0.11	0.05	0.1	0.02	0
		Wbv+A	٨	0.13	0.09	0.04	0.02	0.02	0.08	-0.09	-0.03	-0.01	-0.04	-0.06	0.11	0.05	0.03	0.02	0.15	0.11	0.07	0.13	0
		GG/	$\omega ~({ m cm}^{-1})$	1217.8	1207.2	1196.5	1159.1	1158.3	1130.2	1113.2	1028.8	1028.7	869.1	865.6	830.1	818.8	813.1	810.3	784	772	761.1	748.2	730.5
			γ	0.01	0.01	0.03	0	0	0.02	-0.05	-0.01	-0.01	-0.01	-0.02	0.04	0.01	0	0	0.02	0.09	0.11	0.01	0.01
	culation	GGA	λ	0	0	0.02	0	0	0.01	-0.04	-0.02	-0.02	-0.02	-0.03	0.07	0.04	0.03	0	0.06	0.08	0.12	0	0.01
	Calc)	$\omega (\text{cm}^{-1})$	1203.5	1200.9	1192.3	1153.2	1152.7	1126.2	1122.4	1027	1026.6	869	867.5	831.1	824.7	821.4	811.6	777.2	761.3	742.1	736.7	734
d			۔ ک	0.04	0.04	0.06	0.02	0.01	0.01	-0.09	-0.01	-0.01	0.14	0.10	0.13	0.11	0.00	0.02	0.12	0.00	0.02	-0.02	-0.02
Continue		DA	λ	0.05	0.04	0.05	0.03	0.01	0	-0.08	-0.01	-0.01	0.11	0.08	0.07	0.09	-0.02	0.01	0.06	0.02	0	-0.02	-0.03
TABLE 2: (Ι	$\omega (\text{cm}^{-1})$	1228.2	1222.8	1221	1166	1154.5	1153	1089.4	1030.9	1028.7	886.7	877.1	872.1	864	855.3	853.7	830.8	822.8	802.3	794.4	757

	1)	Satija et	al. (1991)								575	513			451								
	periment (cm ⁻	Deopura and	Gupta (1971)				687				606	510				445		381		360			
	ExJ	Vergoten et	al. (1985)			698					609	513				447		384		362		324	291
			γ	0	0.03	0.03	-0.06	0	0	0	0.07	0.03	0.02	0.05	-0.03	-0.03	0	0.08	0.09	0.00	0.19	0.16	0.05
		Wbv+A	۲	0.01	0.03	0.03	-0.08	0.02	0.01	0	0.11	0.15	0.14	0.08	0	-0.01	0.11	0.26	0.3	0.22	0.35	0.27	0.24
		GGA	ω (cm ⁻¹)	725.2	698.7	698.6	693.4	688.2	650.4	648.2	607.8	512.7	512.2	493.1	442.9	441.8	392	386.7	368.8	365	336.5	333.9	290.8
			4	0.04	0.02	0.01	0	-0.01	0	0	0.04	0.03	0.03	0.04	0.01	0.01	0.06	0.02	0.07	0.03	0.04	0.03	0.07
	culation	GGA	٨	0.06	0.01	0	0	-0.03	0	0	0.03	0.06	0.05	0.04	0	0.01	0.07	0.11	0.15	0.11	0.09	0.08	0.05
	Calc		$\omega (\text{cm}^{-1})$	723.6	696.5	695.9	687.4	685.8	651.5	651.1	599	502.3	501.4	483.2	429.8	428.5	375.5	372.7	349	347.1	329.3	328.4	281.4
q			ر ح	-0.05	0.07	0.05	-0.03	-0.07	-0.02	-0.05	0.15	0.14	0.14	0.13	0.03	0.03	0.27	0.07	0.22	0.2	0.24	0.22	0.3
Continue		DA	٨	-0.04	0.03	0.02	-0.04	-0.06	-0.01	-0.03	0.1	0.11	0.12	0.1	0.04	0.04	0.25	0.07	0.29	0.22	0.26	0.21	0.24
TABLE 2: C		I	$\omega ({ m cm}^{-1})$	754.1	714.8	713.4	692.9	667.8	657.2	654	614	523.4	522.8	499.9	432.2	431.7	403	383	370.8	368.7	344.9	341.5	299.8

International Journal of Energetic Materials and Chemical Propulsion

528

	1)	Satija et	al. (1991)														
	periment (cm	Deopura and	Gupta (1971)		236	173			125	125							
	ExI	Vergoten et	al. (1985)					130			118	96	88	80	57		
			ر ک	0.04	0.44	1.27	1.21	1.04	0.99	0.96	0.93	0.87	1.01	0.78	0.73	0.65	0.45
		Wbv+	λ	0.25	0.87	4.55	4.75	4.39	3.88	4.54	4.53	4.63	5.07	5.57	6.07	6.72	37.78
		GGA	$\omega (cm^{-1})$	288.9	227.4	142.8	141.8	128.5	119.1	114.9	98.8	92.6	81.9	69.7	62.9	49.4	33.5
	_		ر ح	0.05	0.11	0.32	0.29	0.23	0.39	0.39	0.45	0.45	0.53	0.59	0.59		
	ulation	GA	λ	0.11	0.24	1.23	1.22	1.11	2.21	2.76	2.48	2.55	3.31	3.15	2.42		
	Calc	G	$\omega (\mathrm{cm}^{-1})$	279.7	214.6	100.2	93.9	89.9	80.4	72.2	65.4	61	48.2	32.9	18.3	-23.1	-31
led			$\vec{\lambda}$	0.3	0.71	2.95	2.97	2.77	2.75	2.75	2.89	2.86	3.05	3.03	3.04	3.09	3.12
Continu		DA	Х	0.26	0.69	3.16	3.37	2.93	2.43	2.83	2.85	2.7	3.49	3.43	4.46	3.76	15.1
TABLE 2: (Γ	$\omega (cm^{-1})$	299.6	234.8	165.2	151	144	134.3	131.5	125.8	113.9	93.4	68.8	57.8	46.6	37.9

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ne experimental data at ambient condition. γ is the mode Grüneisen parameter [see Eq. (3)] at the gamma point, and $\overline{\gamma}$ is the verage of the mode Grüneisen parameters in the first Brillouin zone [see Eq. (4)]	ABLE 3: Infrared modes of TATB crystal at 0 GPa calculated with the LDA, GGA, and GGA+vdW functionals, along with
verage of the mode Grüneisen parameters in the first Brillouin zone [see Eq. (4)]	ie experimental data at ambient condition. γ is the mode Grüneisen parameter [see Eq. (3)] at the gamma point, and $\bar{\gamma}$ is the
	verage of the mode Grüneisen parameters in the first Brillouin zone [see Eq. (4)]

	(Towns	(1983)							1614	1567				1444					1322	
	oeriment (cm ⁻¹	Deopura and	Gupta (1971)			3317				1618	1571	1549			1447		1416			1322	
	Exp	Vergoten et	al. (1985)	3328	3327	3317	3217	3214		1611	1565	1555			1445		1404			1322	
q. (4)]			Ą	0.01	0.01	0.01	0.02	0.02	0.01	0.05	0.05	0.04	-0.02	-0.03	0.01	0	-0.04	-0.03	0.07	0.02	0
le [see E		Wbv+A	~	0.03	0.02	0.02	0.03	0.03	0.02	0.09	0.08	0.08	-0.02	-0.03	0.03	0.02	-0.1	-0.03	0.09	0.07	0.05
n the first Brillouin zone		GG/	ω (cm ⁻¹)	3389.2	3385.1	3374.5	3278.3	3269.3	3259.7	1561	1558.1	1547.6	1485.6	1481.3	1442.6	1438.2	1391.4	1365.8	1349.5	1301.6	1295.6
he first I	u		ر ح	0.01	0.01	0.01	0.01	0.01	0.01	0.02	0.02	0.01	0	-0.01	0	0	-0.01	-0.02	0.03	0.02	0.01
eters in t	culation	3GA	٨	0.01	0	0.01	0.01	0.01	0	0.02	0.02	0.01	0	-0.01	0	0	-0.04	-0.03	0.02	0.02	0.02
eisen param	Ca	0	$\omega (cm^{-1})$	3373.8	3370.3	3361.5	3267.1	3257.5	3251.7	1548.2	1548	1538.7	1477.8	1475.7	1441.2	1438.1	1394.2	1371.7	1339.5	1289.8	1286.3
e Grüne			4	0.02	0.01	0.01	0.02	0.01	0.01	0.08	0.08	0.07	0.02	0.02	0.02	0	0	0.04	0.07	0.06	0.05
ne mode		DA	λ	0.02	0.01	0.01	0.02	0.02	0.01	0.08	0.07	0.07	0	0	0.02	0.01	0.01	0.03	0	0.06	0.04
iverage of th		ΓI	ω (cm ⁻¹)	3257.1	3251.1	3242.4	3166.8	3155.2	3145.3	1594.6	1593.7	1581.6	1473.4	1464.7	1462.5	1456.5	1398.5	1379.8	1356.1	1335.9	1328.4

International Journal of Energetic Materials and Chemical Propulsion

	(1	Towns	(1983)																		782		
	periment (cm $^{-1}$	Deopura and	Gupta (1971)			1216	1177					1028	880								782		730
	ExJ	Vergoten et	al. (1985)	1229	1221	1219	1173			1050	1033	1028	880					831			783	752	731
			γ	0.03	0.03	0.04	0.03	0.02	0.04	-0.08	-0.01	-0.02	-0.08	-0.1	0.07	0.02	0.01	0	0.09	0.05	0.1	0	-0.01
		Wbv+A	٨	0.06	0.04	0.04	0.02	0.02	0.07	-0.09	-0.02	-0.03	-0.04	-0.06	0.1	0.05	0.06	0.01	0.08	0.06	0.14	0.01	0
		GG,	$\omega (\mathrm{cm}^{-1})$	1203	1198.7	1197.4	1161.9	1159.2	1129.7	1112.2	1029.7	1028	870.1	865.7	826.5	816.5	811.9	810.7	774.3	768.1	752.5	733.6	730.1
			ح،	0	0	0.03	0.01	0.01	0.02	-0.05	-0.01	-0.01	-0.02	-0.02	0.03	0	0	0	0	0.1	0.09	0	0.02
	culation	GGA	٨	0.04	0	0	0	0	0.02	-0.04	-0.02	-0.02	-0.01	-0.03	0.06	0.03	0.03	0	0.01	0.13	0.09	0	0.04
	Calo	0	$\omega (cm^{-1})$	1200.7	1198.3	1196.9	1155.6	1153.2	1126	1121.2	1027.3	1026.4	869.8	867.8	829.3	823.1	817.6	812.1	773.3	751.9	741	734.6	725.3
q			Ą	0.04	0.03	0.06	0.02	0.02	0	-0.09	-0.01	-0.01	0.12	0.1	0.13	0.09	0	0.02	0.09	0.02	0.01	-0.02	-0.03
Continue		LDA	ح	0.05	0.04	0.04	0.03	0.01	0	-0.08	-0.02	-0.02	0.1	0.05	0.07	0.02	0	0.09	0.07	0.06	0	-0.03	-0.03
TABLE 3: ([$\omega (cm^{-1})$	1226.5	1222	1221.3	1167	1155.7	1152.8	1088.8	1028.4	1026.1	884.7	872.9	867.9	857.6	854.3	845.3	829.3	818	800	793	755

	(1	Towns	(1983)																				
	periment (cm $^{-1}$	Deopura and	Gupta (1971)				700						510			445							
	Exj	Vergoten et	al. (1985)	725		708	697				606	567	517		448	446				365			
			γ	0.01	0.02	0.02	-0.08	-0.01	0.03	-0.02	0.06	0.03	0.02	0.04	-0.04	-0.05	-0.01	0.07	0.08	0.01	0.2	0.18	0.05
		Wbv+A	λ	0.03	0.04	0.04	-0.09	-0.01	0.07	-0.01	0.11	0.14	0.13	0.11	-0.02	-0.03	0.15	0.25	0.24	0.24	0.35	0.25	0.24
		GG/	$\omega (\text{cm}^{-1})$	723.2	869	697.3	693.3	683	655.5	647.5	607	513.3	511	492.3	441.8	441	393.4	385.4	367.4	366.6	338.3	334.9	291.1
			-21	0.05	0.01	0.01	-0.01	-0.01	0.01	0	0.04	0.03	0.03	0.04	0.01	0.01	0.06	0.02	0.07	0.03	0.05	0.03	0.07
	culation	GGA	λ	0.05	0.01	0.01	-0.01	0	0.01	0	0.03	0.05	0.05	0.04	0.01	0.01	0.07	0.06	0.13	0.09	0.1	0.08	0.08
	Calc		$\omega (\text{cm}^{-1})$	720.7	695.8	695	686	682.2	653.3	650.7	598.1	502.3	501.3	481.6	428.7	428.1	376.8	372.2	348.7	347.3	330	329.1	282
q			-21	-0.04	0.07	0.05	0	-0.08	-0.01	-0.03	0.16	0.14	0.13	0.12	0.02	0.02	0.27	0.05	0.17	0.13	0.24	0.24	0.31
Continue		DA	γ	-0.04	0.03	0.03	0	-0.07	0.01	-0.03	0.08	0.11	0.1	0.08	0.04	0.03	0.25	0.02	0.18	0.2	0.28	0.19	0.25
TABLE 3: (Ι	$\omega (\text{cm}^{-1})$	754.8	714	713.1	696.2	667	662.2	655	614.5	523.1	521.8	499	431.3	430.9	403.4	379.6	367.8	365.7	347.6	342.4	300.2

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International Journal of Energetic Materials and Chemical Propulsion

532

1				1	1		1	r	1	1		1	[
	(Towns	(1983)											
	periment (cm $^{-1}$	Deopura and	Gupta (1971)											
	ExJ	Vergoten et	al. (1985)	290		135	122	114				86	72	58
			2	0.04	0.51	1.08	1.03		0.98	0.86	1.14	0.87	0.63	0.56
		Wbv+	λ	0.26	0.87	4.44	4.89	4.13	5.25	4.62	5.12	4.9	5.57	7.8
		GGA	$\omega (\text{cm}^{-1})$	290.2	227.4	134.1	128.1	119.4	108.5	94.3	81.9	76.8	57.3	42.7
			2	0.06	0.14	0.25	0.31	0.31	0.46	0.45	0.51	0.58	0.58	0.6
	lation	GA	٨	0.07	0.24	0.96	1.16	1.36	0.88	1.79	2.26	3.1	3.26	12.8
	Calcu	G	$\omega (\mathrm{cm}^{-1})$	281.3	215.2	93.3	88.4	85.3	70.8	63.2	60.6	38.5	35.5	20.3
led			γ	0.3	0.74	2.8	2.82	2.72	2.95	2.91	2.96	3.05	3.1	3.03
Continu		DA	\prec	0.24	0.74	3.03	2.67	2.32	2.1	2.99	3.56	3.25	4.25	3.08
LABLE 3: (Γ	$\omega (\text{cm}^{-1})$	299.3	235.6	146.6	139.4	132.9	114.6	110.3	97.1	92.4	73	63.6

Volume 14, Number 6, 2015

$$\gamma_{i,q} = -\frac{\partial \ln \omega_{i,q}}{\partial \ln V} \tag{3}$$

where *i* is the mode index and *q* is the wave vector in the reciprocal space. (Note that there is no thermal expansion within the harmonic approximation because $\gamma_{i,q} = 0$.) The calculated mode Grüneisen parameters in TATB crystal have the same features as those in β -HMX A (Matsui et al., 2012). γ_i values are given in Tables 2 and 3 for q = 0 (gamma point), which clearly depend on the mode frequency. For high-frequency modes, γ_i is close to zero because the lengths of strong molecular bonds in the molecular crystal are not sensitive to the volume change. On the contrary, low-frequency modes usually have γ_i values much larger than 1. Tables 2 and 3 also list the integration of γ_i over the first Brillouin zone:

$$\bar{\gamma}_i = \frac{\int \gamma_{i,q} dq}{\int dq} \tag{4}$$

High-frequency modes in some way are like Einstein modes, and their γ_i and $\overline{\gamma}_i$ values are close to each other, since their frequencies are almost independent of wave vector q. For low-frequency modes, LDA and GGA calculations exhibit vastly different behaviors. γ_i and $\overline{\gamma}_i$ calculated by LDA are not much different, which means that the gamma-point (q = 0) value γ_i is a good representation of the integration of $\gamma_{i,q}$ over the first Brillouin zone. In contrast, $\overline{\gamma}_i$ is only about $1/4 \sim 1/2$ of γ_i in the GGA calculations. Therefore, thermodynamic properties derived from GGA calculations using only the gamma point are not reliable. The vdW correction increases $\overline{\gamma}_i$ only slightly and does not change the major feature that $\overline{\gamma}_i$ is far smaller than γ_i .

In GGA calculations, we obtain two imaginary modes at gamma point at 0 GPa, whose frequencies appear as negative values in Table 2. At higher pressures, the imaginary modes disappear. The imaginary modes from GGA cause a problem in the calculation of thermodynamic properties using QHA, see Eq. (2). However, both modes are imaginary only for a narrow range of q values around the gamma point. The contribution of the imaginary frequencies to the phonon density of states (PDoS) is negligible. Furthermore, these imaginary modes may become stable after including phonon-phonon interactions, since the triclinic TATB is stable at ambient condition. Therefore, it is reasonable to calculate thermodynamic properties of TATB using QHA by ignoring the two imaginary frequencies. Although the proportion of PDoS of the imaginary frequencies to that of all frequencies is very small, it increases with the volume. Consequently, the calculated thermodynamic properties become less reliable above a certain volume. Note that the zero-point motion [second term in Eq. (2)] increases the volume by about 3% but has a small effect on thermodynamic properties. By ignoring this term, we can thus obtain more reliable thermodynamic properties. Accordingly, the volume difference between the static and 300 K GGA calculations arises solely from the temperature contribution; see Table 1.

Thermal expansion of TATB crystal calculated with the LDA, GGA, and GGA + vdW functionals is compared with experimental data in Fig. 3. The GGA and LDA



FIG. 3: (a) Thermal expansion of TATB at 0 GPa. Experimental data are obtained by fitting the volume vs temperature data by Kolb and Rizzo (1979). (b) Relative volume variation with temperature, where V(T) is the volume at temperature T. Experimental data are taken from Kolb and Rizzo (1979). Results from the LDA, GGA, and GGA+vdW calculations are shown.

calculations predict nearly the same thermal expansion below 150 K. Above 150 K, the thermal expansion in GGA increases rapidly and exhibits a superlinear behavior, which is likely due to the omission of anharmonic effects in QHA (Matsui et al., 2012). The experimental data in Fig. 3, which are obtained by fitting the measured volume vs temperature data in Kolb and Rizzo (1979) are located between the LDA and GGA results. As mentioned in Sec. III-A, the compression ratio among a, b, and c calculated by GGA is 1:0.94:9.7 below 0.5 GPa. Consequently, the linear coefficient of the thermal expansion of the c axis is almost 10 times that of the b axis according to the constrained QHA (Carrier et al., 2007), which assumes that the lattice parameters only depend on the volume. [This assumption was shown to work well for MgSiO₃ (Carrier et al., 2007)]. The result is consistent with the experimental data, which indicate that the relative expansion of the c axis is about 12 times that of the b axis. This suggests that the constrained QHA can also be applied to molecular crystals such as TATB. The LDA calculation predicts that the relative expansion of the c axis is about 6 times that of the b axis, since the compression ratio along a, b, and c from LDA is 1:1.04:5.92 below 1 GPa, which is slightly smaller than the experimental data. Both the LDA and GGA results show that the a and b axes have almost the same linear coefficients of thermal expansion, which does not agree with the Kolb and Rizzo (1979) experimental results that the relative expansion of the b axis is about 3 times of that of the a axis but is consistent with the recent measurement by Sun et al. (2010). The vdW correction makes the thermal expansion of TATB much smaller than the GGA result and worsens the agreement with the experimental data (Kolb and Rizzo, 1979).

Similar to the cases of other energetic molecular crystals such as β -HMX (Matsui et al., 2012), the phonon contribution to the equilibrium volume is remarkably large in

TATB compared to nonmolecular crystals. As shown in Table 1, the equilibrium volume increases about 7% (5%) when phonon effects are included with GGA (LDA). Based on our results on TATB and β -HMX and similarly large thermal expansion of other energetic materials such as RDX (Cady, 1972), PETN (Cady, 1972), and NTO (Bolotina et al., 2003), observed experimentally, we expect that the large vibrational contribution to the volume is a common feature of all energetic molecular crystals. This is in contrast to the most nonmolecular crystals (Li et al., 2007; Mounet and Marzari, 2005; Pavone et al., 1993; Sun et al., 2008; Wentzcovitch et al., 2010; Wu and Wentzcovitch, 2007; Wu et al., 2008; Yu and Wentzcovitch, 2006), where the vibrational contribution to the volume is only $1 \sim 2\%$. Consequently, the common practice of evaluating the exchange-correlation functional by comparing the static DFT calculation results with room temperature experimental data is problematic for molecular crystals like the energetic materials. The LDA functional is usually not preferred, because it significantly underestimates the volume of most molecular crystals with static calculation. However, when the vibrational effects are included, the discrepancy between the volume predicted by LDA and the experimental value is considerably reduced. For TATB, the discrepancy between LDA and the experimental volume decreases from -9.6% to -4.6% when vibrational effects are included (see Table 1). Therefore, contrary to the common belief, we find that LDA is a good exchange-correlation functional for energetic molecular crystals like TATB and HMX. The equilibrium volume from GGA+vdW increases about 2.5% when phonon effects are included, which is slightly smaller than previous works (Landerville et al., 2010).

The equation of state and bulk modulus of TATB at 300 K calculated with LDA, GGA, and GGA+vdW functionals are compared with experimental data in Fig. 4. Overestimation of the volume by GGA decreases with the pressure. The vdW correction reduces the volume overestimation by GGA (by 32% at 0 GPa and 4% at 10 GPa), and the resulting equation of state with the vdW correction agrees well with the experimental data. This indicates that the bulk modulus with the vdW correction is quite accurate, which is confirmed by the good agreement between the calculated result and the experimental data (Olinger and Cady, 1976; Stevens et al., 2008) at 0 GPa; see Table 4. As shown in Fig. 4(b), the bulk modulus from GGA is much smaller than the experimental data even at high pressures, and the bulk modulus from LDA is almost identical to that from GGA+vdW. This means that the LDA accurately predicts the bulk modulus in the entire pressure range. Thus the error in volume by the LDA calculation is invariant with the pressure. The error in volume from the LDA is 4.6% at 0 GPa (see Table 1). By multiplying the LDA volume by a constant factor of 1.048, we get an equation of state that agrees well with the experimental data in the entire pressure range. This simple recipe for obtaining an accurate equation of state has been shown to work for β -HMX as well, where we obtain a highly accurate equation of state of β -HMX by multiplying the volume from LDA by a constant factor of 1.06. (The LDA error in the volume at ambient condition is 6% for β -HMX.) We expect that this is also true for other energetic



FIG. 4: Pressure dependence of (a) volume and (b) isothermal bulk modulus. Results from the LDA, GGA, and GGA+vdW calculations are compared with the experimental data by Kolb and Rizzo (1979) (solid squares) and Stevens et al. (2008) (open circles).

molecular crystals such as RDX, PETN, and NTO. Thus, an accurate equation of state for an energetic molecular crystal at broad temperature and pressure range can be obtained by multiplying the volume calculated within the LDA by a constant factor (1+error), where the *error* is the relative volume difference between the LDA and experimental results at ambient condition. In contrast, the equation of state of energetic molecular crystals from the GGA+vdW method can only apply to the narrow temperature around room temperature because of its significant underestimation of thermal expansion.

As shown in Fig. 5, the LDA also reproduces the experimental heat capacity at constant pressure C_P of TATB very well (Baytos, 1979), as does GGA. However, the vdW correction largely underestimates C_P . Overall, the GGA is good at describing the thermal expansion α and C_P , but fails to describe the isothermal bulk modulus K_T and the equilibrium volume V. On the other hand, GGA+vdW is good at describing K_T and V, but fails for α and C_P . The LDA accurately describes all thermodynamic properties— K_T , C_P , α , V—with a slight underestimation of $V \sim 5\%$), which can be corrected easily by multiplying with a constant factor as mentioned above. Since the thermal Grüneisen parameter γ_{th} is related to these properties through the relation

TABLE 4: Thermodynamic properties of TATB crystal at ambient condition, where α is the thermal expansion, C_P and C_v are the heat capacities at constant pressure and constant volume, respectively, γ_{th} and V are the thermal Grüneisen parameter and the equilibrium volume, and K_T and K'_T are the isothermal bulk modulus and its pressure derivative

$\frac{\alpha}{(10^{-5}/\mathrm{K})}$	$\frac{C_P}{(\mathbf{J/mol/K^{-1}})}$	$\frac{C_V}{(\mathbf{J/mol/K^{-1}})}$	γ_{th}	V (Å ³)	<i>К</i> _Т (GPa)	K_T'	
20.68	276.60	257.32	1.20	422.3	11.8	12.7	LDA
43.40	274.35	260.44	0.41	591.58	1.38	12.2	GGA
5.22	259.92	289.52	0.35	442.88	12.85	12.9	GGA+ vdW
30.2 Kolb and Rizzo (1979)	280.69 Pastine and Bernecke (1974)			442.9 Olinger and Cady (1976)	13.6 Stevens et al. (2008) 16.7 Olinger and Cady (1976)	12.4 Stevens et al. (2008) 5.7 Olinger and Cady (1976)	Experi- ment



FIG. 5: (a) Heat capacity and (b) thermal Grüneisen parameter at 0 GPa. Results from the LDA, GGA, and GGA+vdW calculations are compared with the experimental data by Pastine and Bernecke (1974).

$$\gamma_{th} = \frac{\alpha V K_T}{C_V} \tag{5}$$

where C_V is the heat capacity at constant volume, γ_{th} estimated with the LDA should be reasonable as well. As shown in Fig. 5, the GGA gives γ_{th} much smaller than the LDA value, and the vdW correction only slightly changes the GGA value. Why γ_{th} from the

International Journal of Energetic Materials and Chemical Propulsion

GGA is smaller than from the LDA is understood as follows: γ_{th} is a weighted average of the mode Grüneisen parameter $\bar{\gamma}_i$, and at low temperatures, low-frequency modes provide dominant contributions to γ_{th} . As listed in Tables 2 and 3, $\bar{\gamma}_i$ for low-frequency modes calculated with the GGA are significantly smaller than those obtained from the LDA, which makes the GGA estimate of γ_{th} very small. Though the GGA considerably underestimates $\bar{\gamma}_i$, the GGA can still describe α well, because it also underestimates VK_T at the same time to cancel out this error. (K_T is only about a tenth of the experimental value; see Tables 1 and 4.) The vdW correction improves VK_T dramatically but changes $\bar{\gamma}_i$ only slightly. As a result, the vdW correction worsens the α calculated with the GGA. In order to describe thermodynamic properties correctly, the vdW correction must increase $\bar{\gamma}_i$ significantly in the GGA calculation. Note that all dispersion coefficients in the empirical vdW method adopted here are fixed independent of the electron density distribution. In reality, the local electron density varies from atom to atom, and this should affect dispersion coefficients. Although this effect may change the phonon frequency only slightly, it may cause a dramatic change in $\bar{\gamma}_i$, i.e., the volume dependence of the phonon frequency. Some vdW methods (Dion et al., 2004; Tkatchenko and

4. CONCLUSION

better estimates of all thermodynamic properties.

We have investigated the effect of the exchange-correlation functional on the vibrational and thermodynamic properties of TATB molecular crystal. Our main finding is that the LDA is a good exchange-correlation functional for predicting these properties. The same conclusion was drawn previously for β -HMX (Matsui et al., 2012). Previous works also indicated that LDA is much better than GGA for graphite and the adsorption of adenine on graphite (Hasegawa and Nishidate, 2004; Ortmann et al., 2005). We thus expect that this is likely to be a common feature among other energetic materials or even among broader molecular crystals.

Scheffler, 2009) are being developed to incorporate this effect, which would provide

The GGA describes the thermal expansion coefficient α of TATB and β -HMX (Matsui et al., 2012) quite well, which is remarkable given the fact that the GGA dramatically underestimates their bulk moduli K_T . Our phonon calculations reveal that this is due to a cancellation of errors, i.e., the GGA also underestimates the mode Grüneisen parameters $\bar{\gamma}_i$, resulting in a small thermal Grüneisen parameter γ_{th} . The underestimations in γ_{th} and K_T cancel each other, and thus the GGA predicts α quite well. The empirical vdW correction resolves the underestimation of K_T by the GGA but only slightly increases γ_{th} , making α from the GGA+vdW too small. When compared with the GGA and GGA+vdW, the LDA satisfactorily describes all thermodynamic properties, including α , K_T , C_P , and γ_{th} .

Because molecular crystals like TATB and β -HMX (Matsui et al., 2012) have similar γ_{th} and much smaller K_T compared with other types of crystals, α of molecular

Volume 14, Number 6, 2015

crystals are usually several times larger than that of nonmolecular crystals. Accordingly, the volume of a molecular crystal changes significantly with temperature. Therefore, for molecular crystals, it is essential to include phonon contributions to the volume when evaluating the quality of the exchange-correlation functional. The GGA calculations, which already overestimate the volume of TATB by 27% in static calculations, become worse after including phonon contributions. In contrast, the LDA results become much better after including phonon contributions, with the volume deviation from the experimental value of only 4.6% at room temperature. Furthermore, the error in volume from the LDA does not change with pressure, since the LDA accurately predicts the bulk modulus of TATB over a wide range of pressure. Consequently, we can simply obtain a highly accurate equation of state of TATB from the LDA calculations.

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Volume 14, Number 6, 2015

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